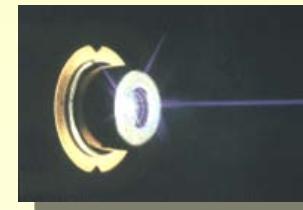


Inhomogeneous electron distribution in InN nanocolumns

Jaime A. Segura Ruiz

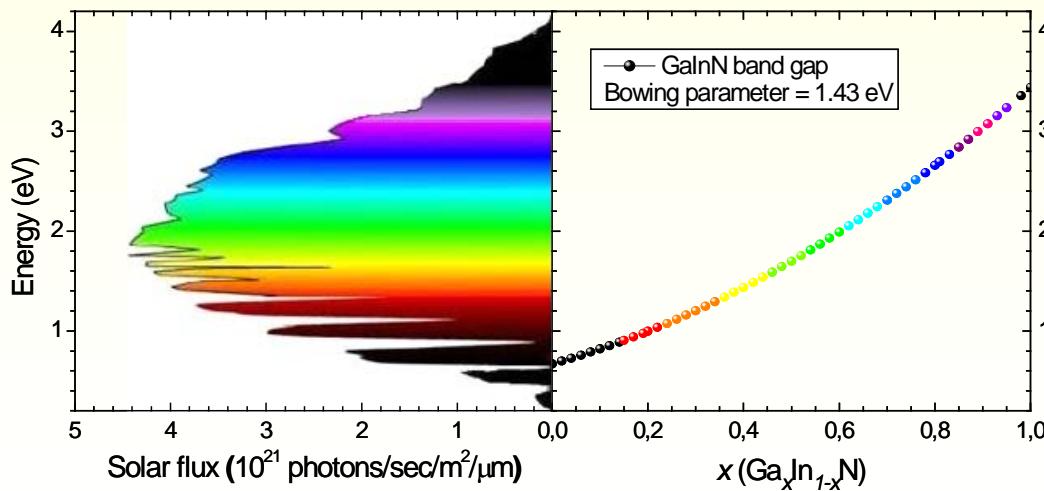
Electronic, transport and optical properties of low-dimensional systems
WS10-ETOLDs.
Valencia, May 31 of 2010

Nitride semiconductors: GaN, AlN, InN

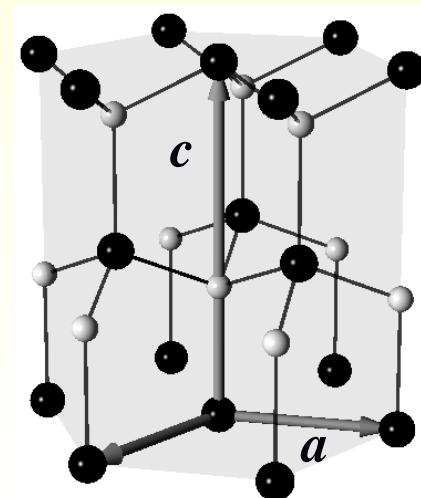


Blue Laser Disc
Write-Once
100GB 4層

Bandgap of $\text{Ga}_{1-x}\text{In}_x\text{N}$ – Solar spectrum
High efficiency solar cells.



Wurtzite structure
Non-centrosymmetric



Polar and non-polar
surfaces

Indium nitride

Low crystalline quality layers.
Bandgap (E_G): 1.9 eV

2002

High crystalline quality.
Reassigned E_G : 0.67 eV.

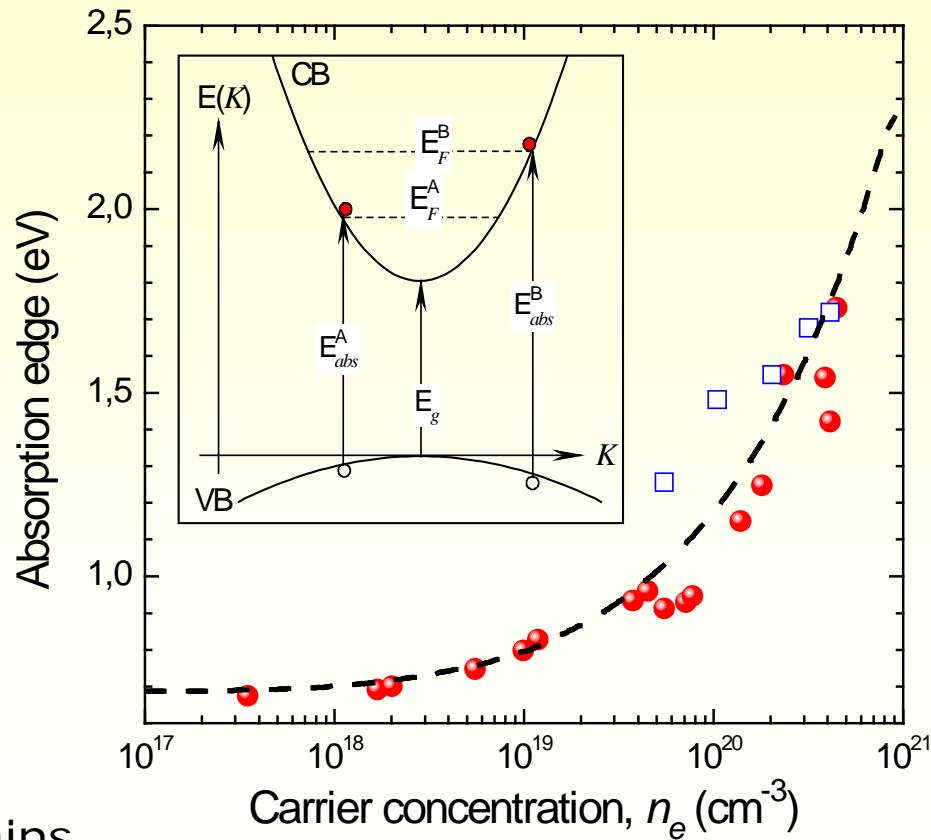
High density of electrons
(donors)



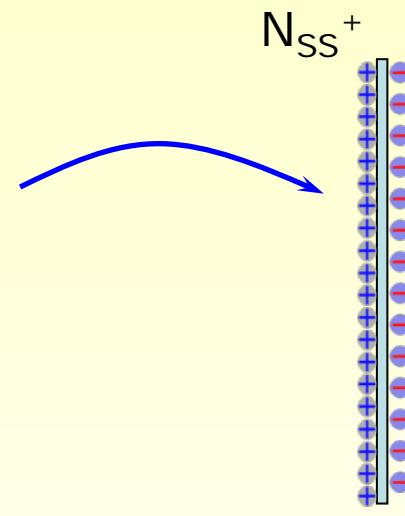
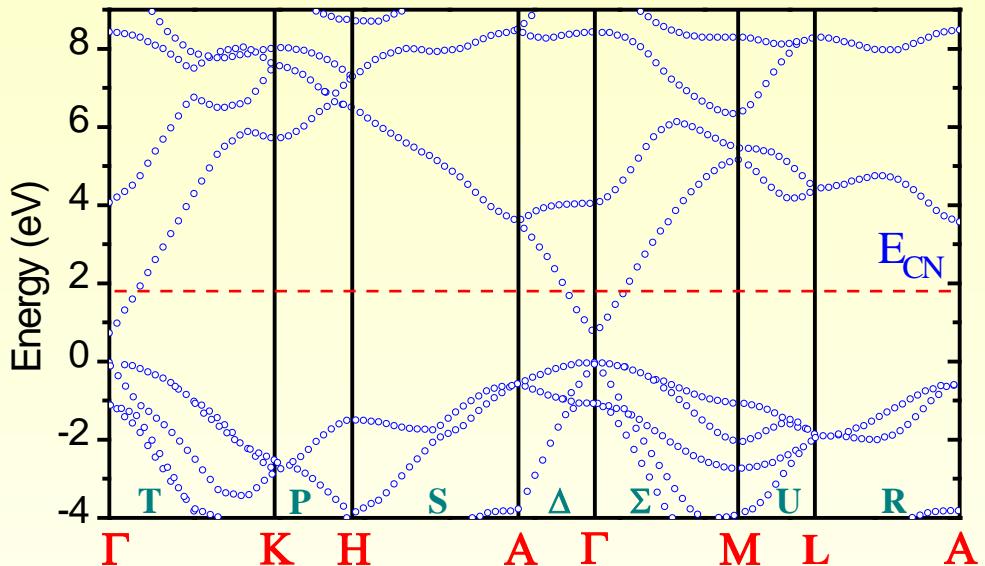
Electron effective mass very
small

Large shift of the absorption
edge (Burstein-Moss).

Free electron concentration remains
high: $n_e > 3 \times 10^{17} \text{ cm}^{-3}$

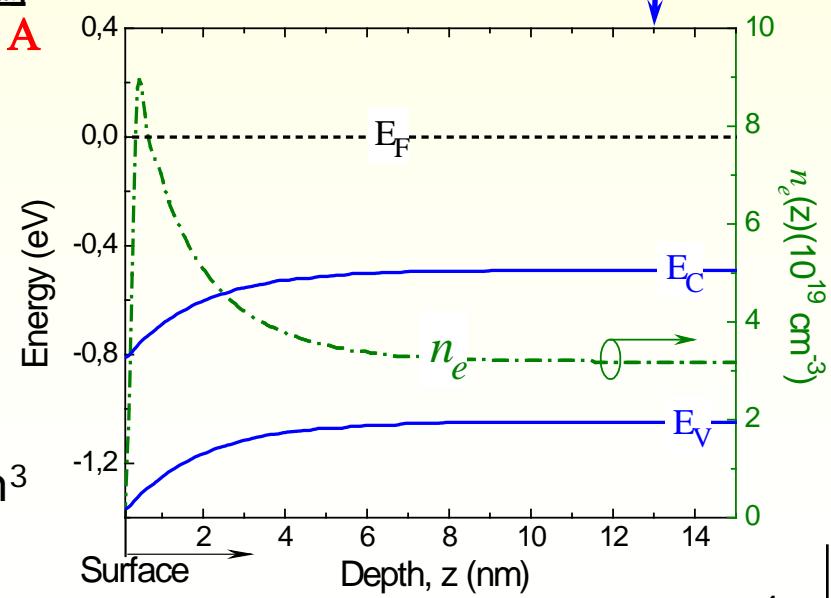


Surface electron accumulation layer.

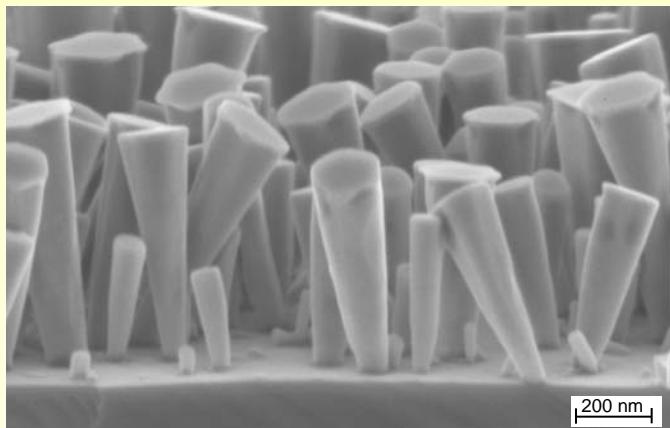


Polar surfaces: Intrinsic

Non-polar surfaces {
 Intrinsic¹
 Extrinsic²
 No accumulation³



Nanocolumns (NCs)

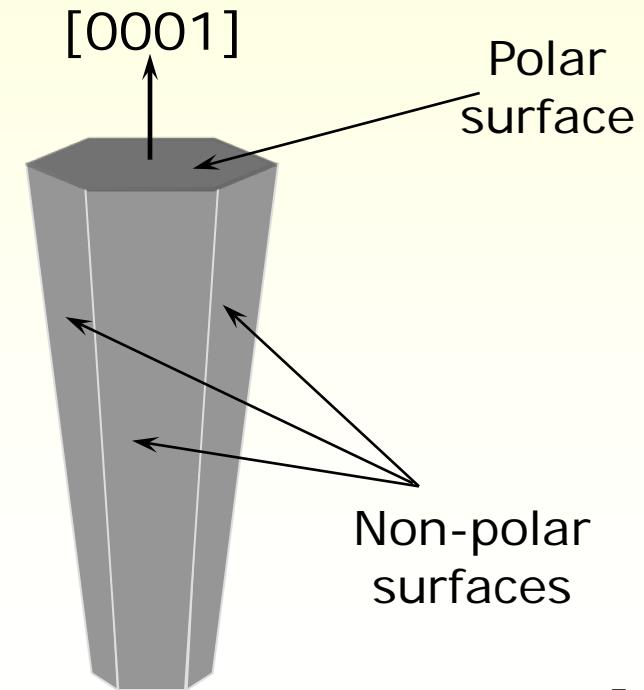


Why NCs?

- Higher surface/volume ratio.
- Lower density of dislocations.
- Increase radiative recombination.

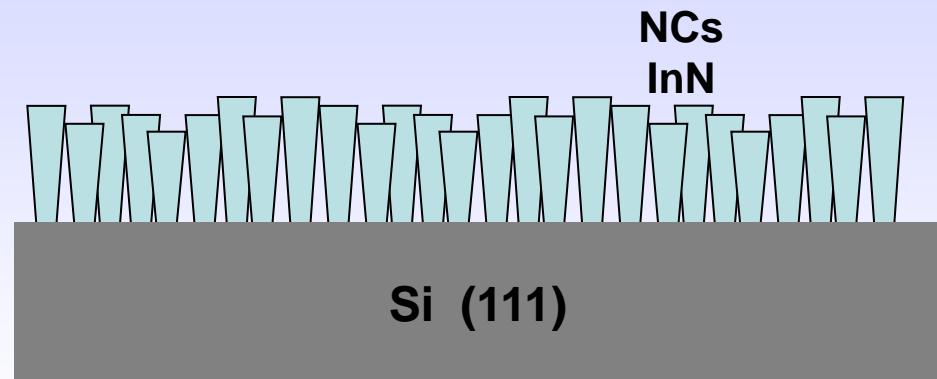
NCs grow spontaneously along c-axis.
Most surface is non-polar.

Non polar surfaces properties can be studied in more detail.



InN NCs: samples

- Growth: Plasma-assisted MBE.
- N_2 -rich conditions
- Growth time: 300 m.

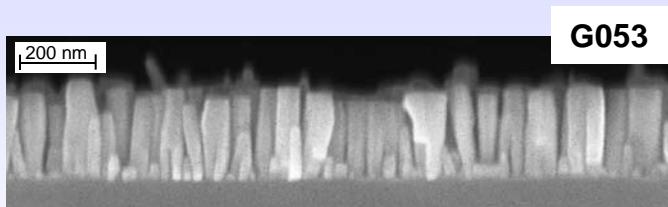


T_s , Indium beam equivalent pressure (BEP), N_2 -flux: variables.

Sample	T_s ($^\circ C$)	In-BEP (mbar)	N_2^{flux} (sccm)	P_{RF} (W)
G053	400	3.0 ± 10^{-8}	2.0	500
G071	475	3.0 ± 10^{-8}	2.0	500
G047	500	3.0 ± 10^{-8}	2.0	500
G041	500	1.5 ± 10^{-8}	1.5	400
G044	500	3.0 ± 10^{-8}	1.5	450
G136	475	3.0 ± 10^{-8}	1.5	450

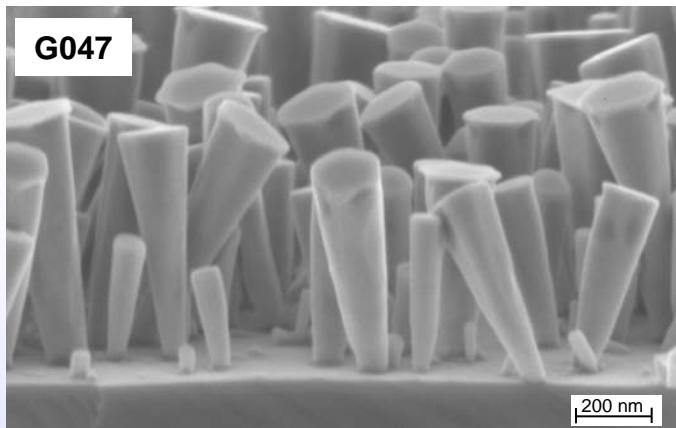
Morphology

Growth conditions strongly affect NCs morphology.



$T_s = 400^\circ\text{C}$: Coalescence

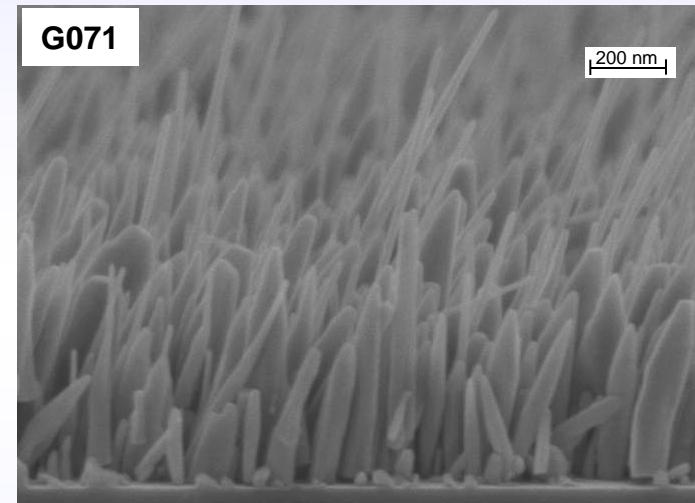
$T_s = 500^\circ\text{C}$: Baseball-bate shape



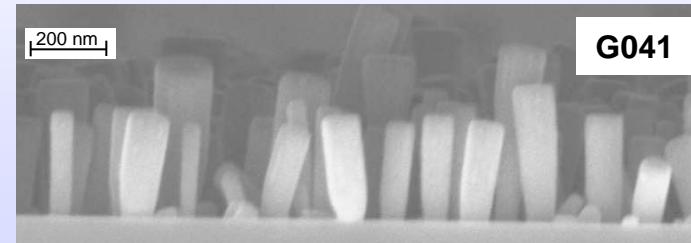
Diameters: 80-150 nm (30 nm)

Heights: 200-600 nm (1500 nm)

$T_s = 475^\circ\text{C}$: Tapering - bimodal

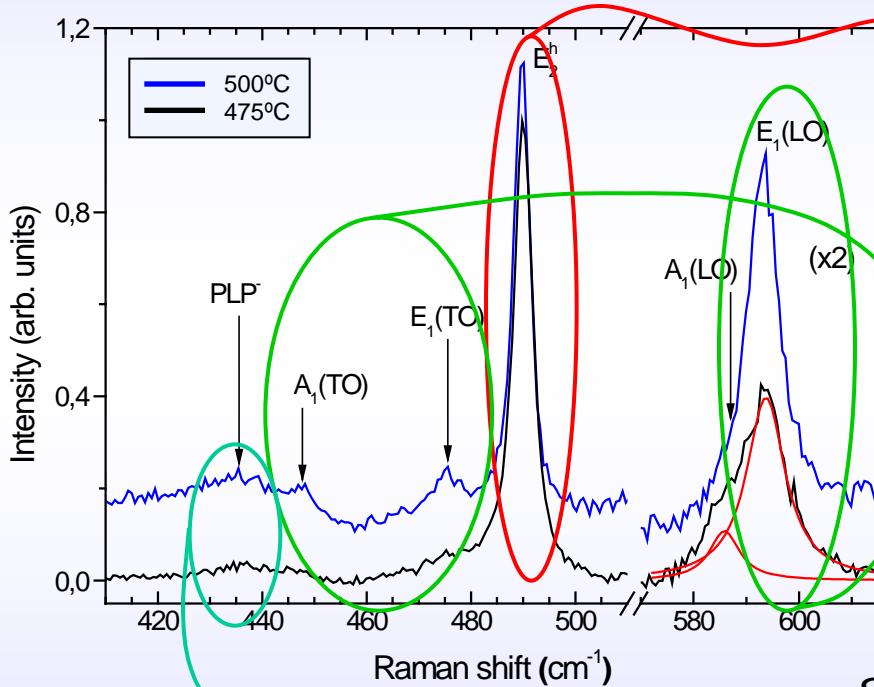


$T_s = 500^\circ\text{C}$: Homogeneous NCs



Raman scattering spectroscopy

Allowed modes: $A_1(\text{LO}) - E_2^h$



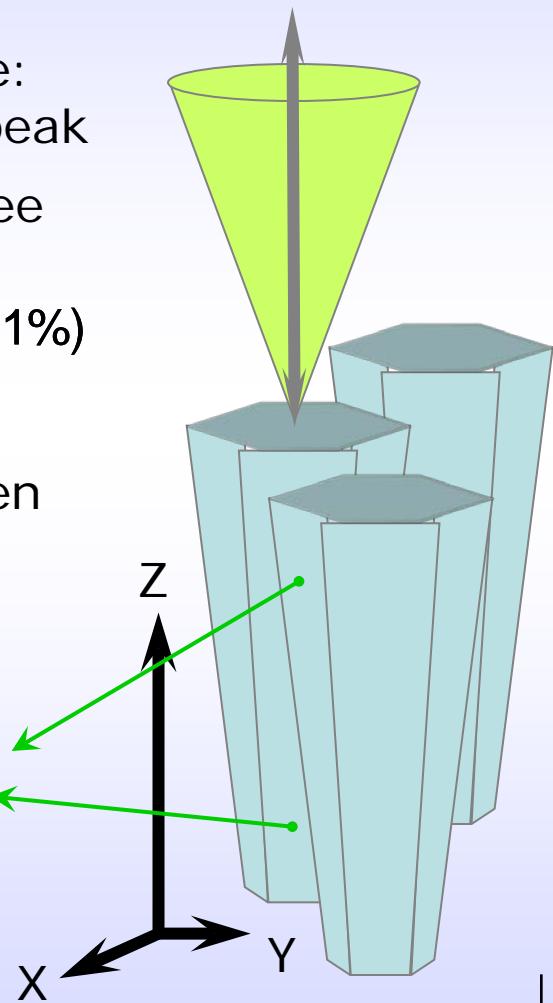
Plasmon - LO phonon coupled mode (PLP⁻)

E_2^h mode:
narrow peak
Strain free
NCs
($\epsilon_{xx} < 0:1\%$)

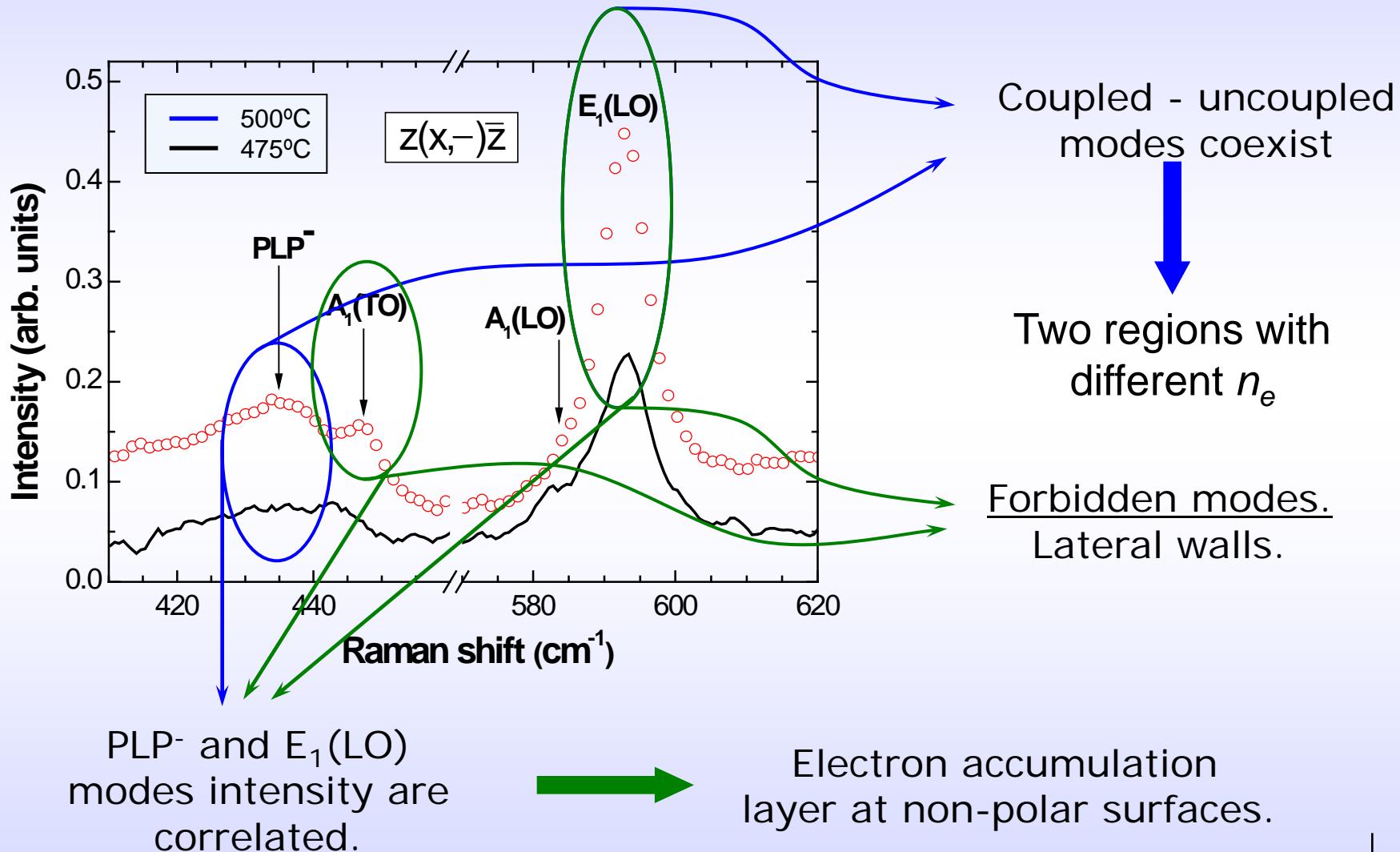
Forbidden modes

Scattering at lateral walls.

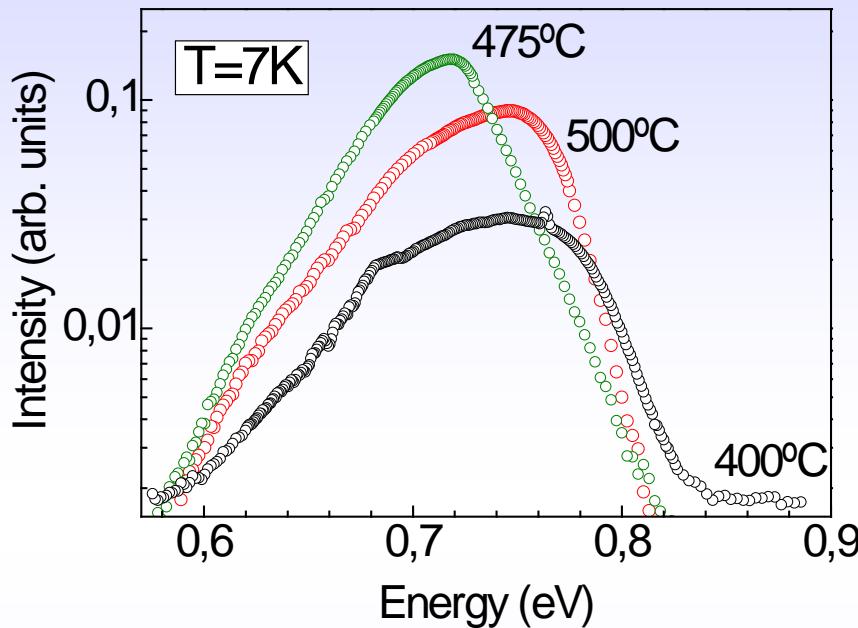
Backscattering
 $Z(X, —)-Z$



Raman scattering spectroscopy

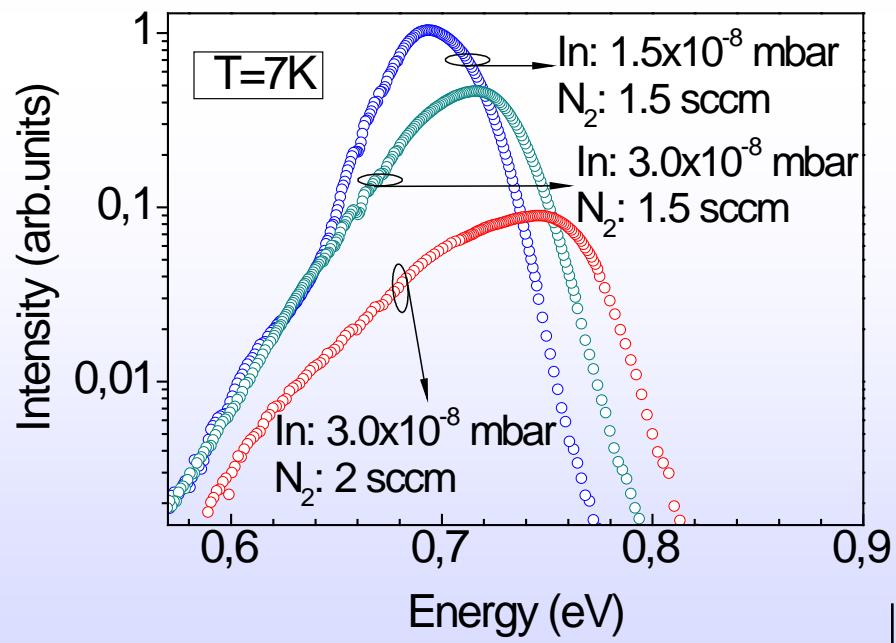


Photoluminescence: growth conditions

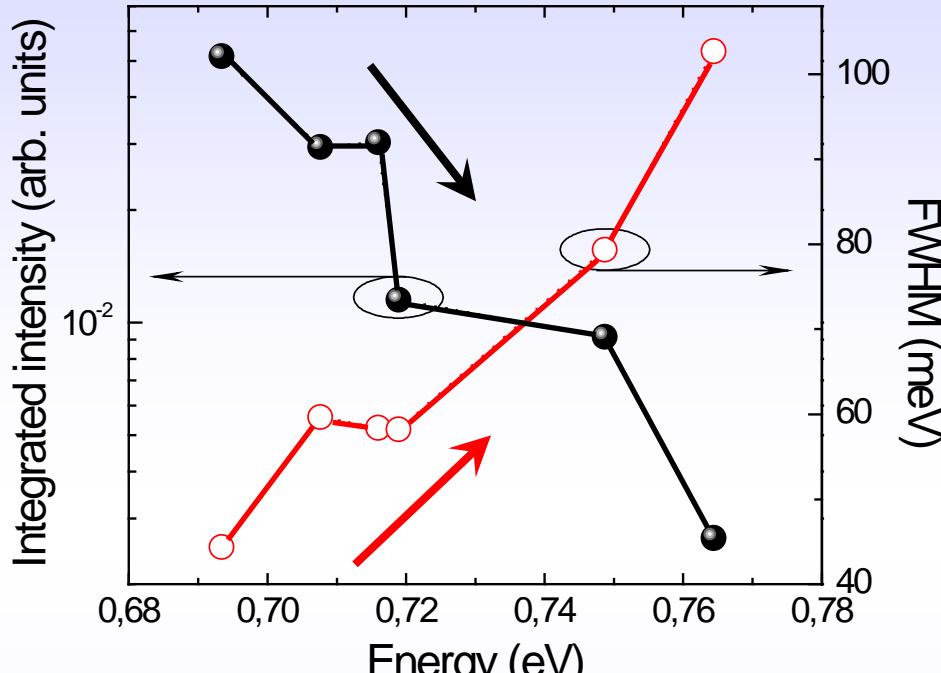


In-BEP and N₂ –flux effect

T_s effect



Photoluminescence



$$I(T) = \frac{I_0}{1 + A \exp(-E_1/kT) + B \exp(-E_2/kT)}$$

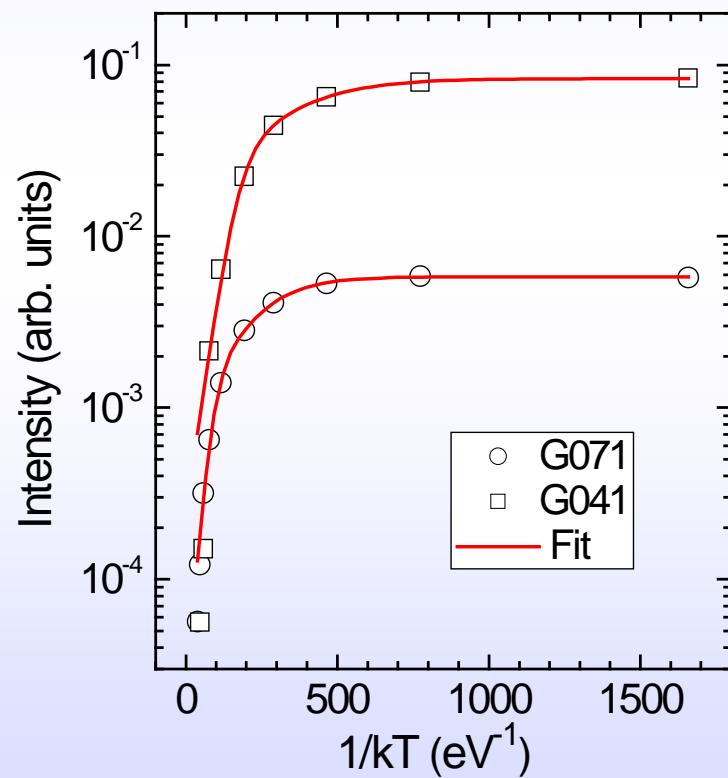
Two non-radiative channels.

E_1 : 5-10 meV. Shallow acceptors

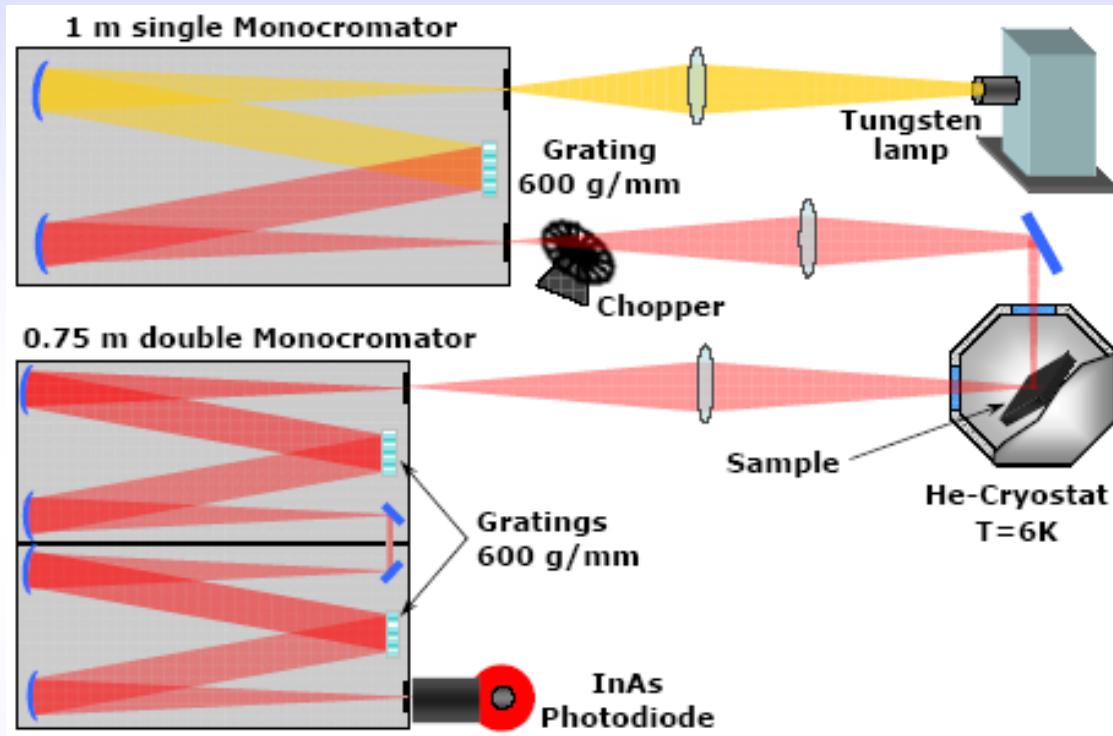
E_2 : 40-50 meV. Deep acceptors

Energy, intensity and FWHM are correlated.

Temperature dependence



Photoluminescence excitation (PLE)

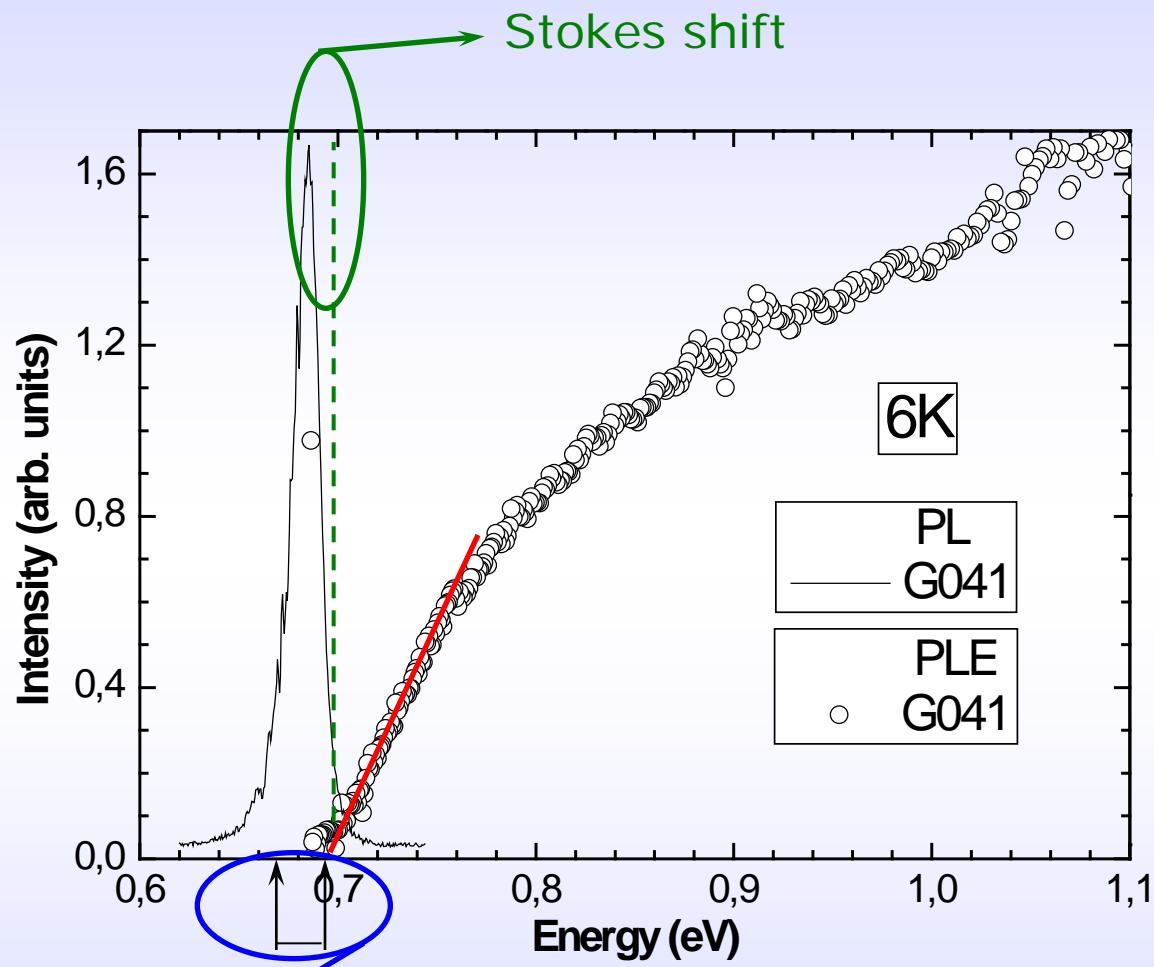


- 10K.
- Excitation: Halogen lamp + monochromator.
- Detection: N_2 -cooled InAs Photodiode.
- Spectral resolution: $<1.5 \text{ meV}$

PLE: characteristics

Line-shape similar to bulk material.

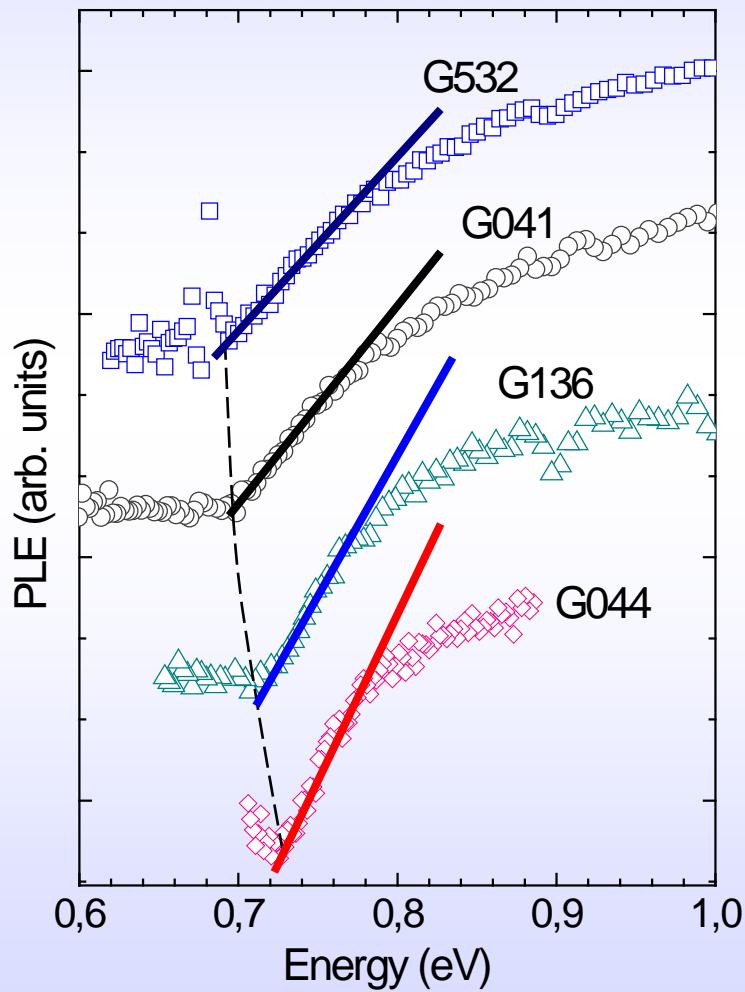
Featureless – High electron concentration



E_{abs} higher than InN bandgap

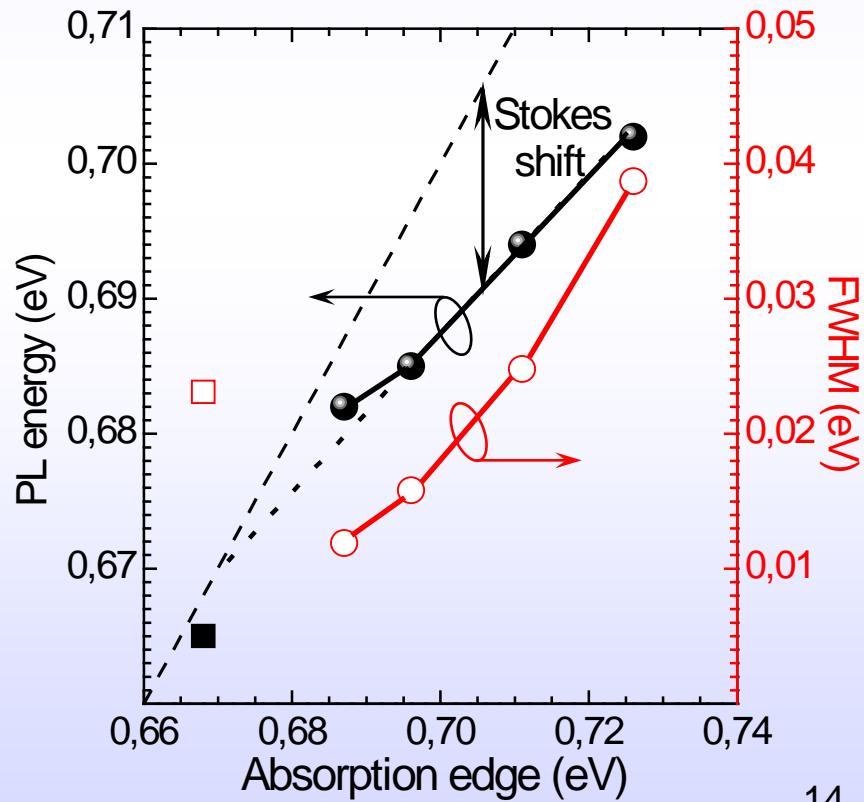
Accumulation layer: growth conditions

PLE is different for each sample.



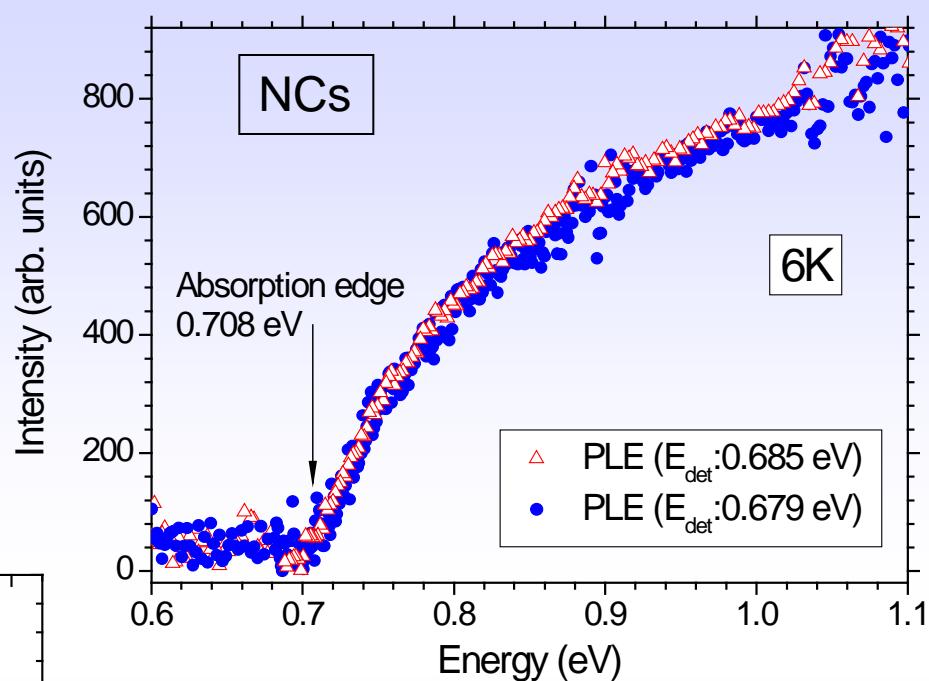
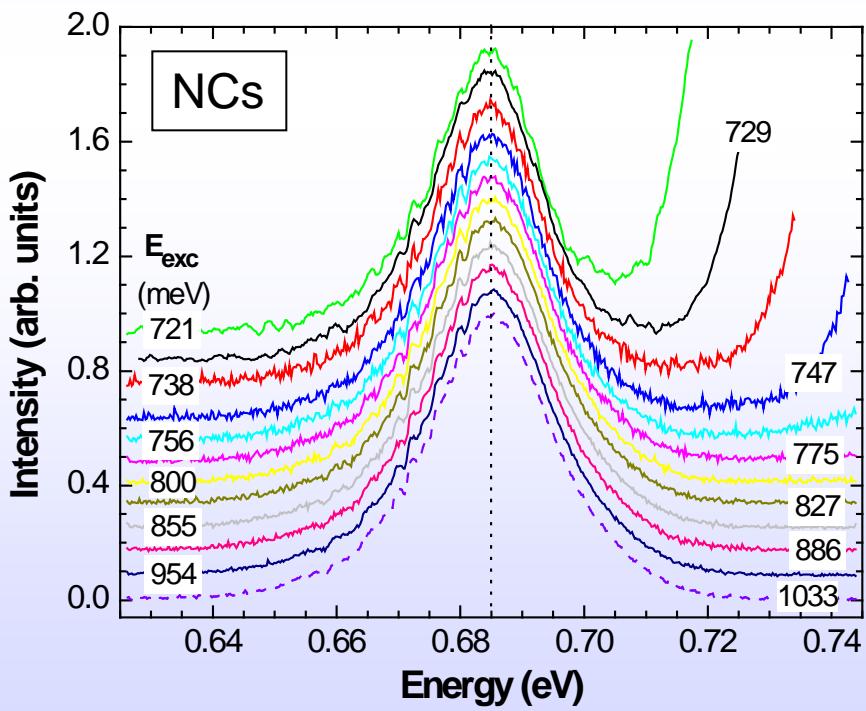
Energy and FWHM of the PL peak increase as E_{abs} increase.

PL peak energy does not follow E_{abs} variation.



Broadening

PL does not depend on E_{exc} .



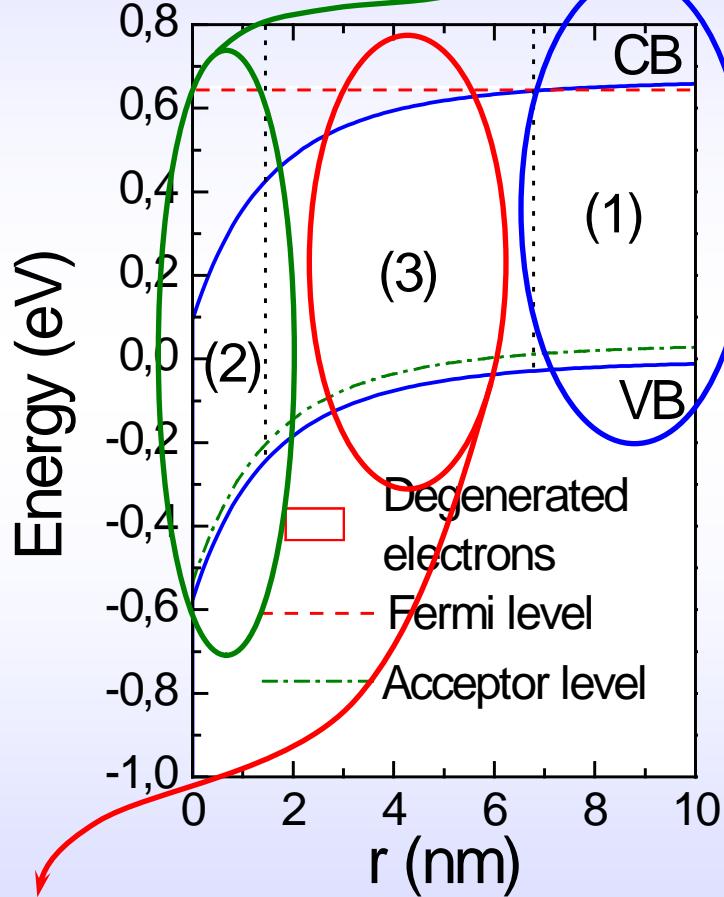
E_{abs} does not depend on E_{det} .

**Homogeneous
broadening**

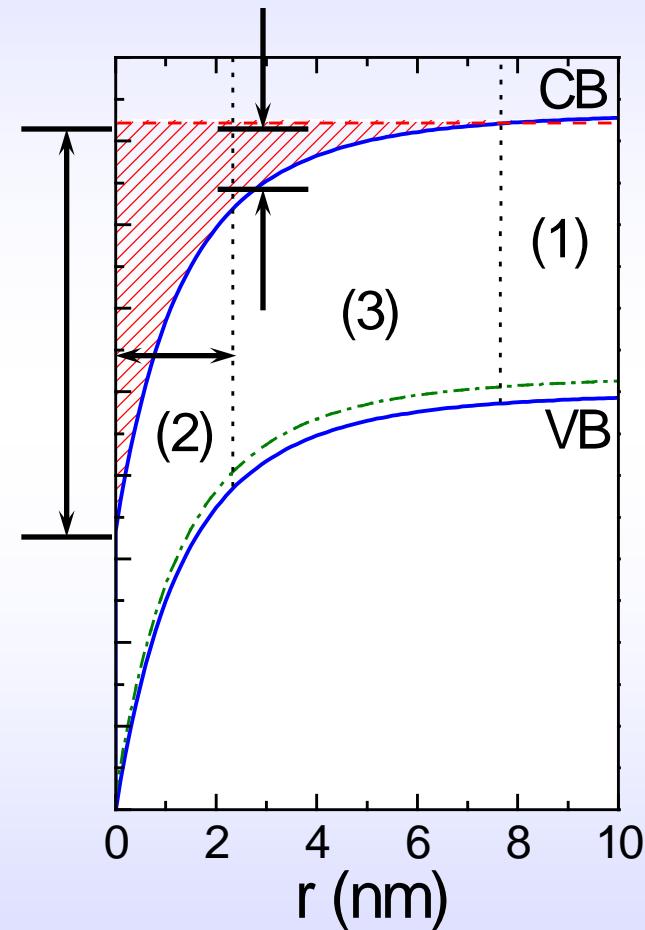
Schematic NC image

Hole depletion

Low PL efficiency



Lower e^- concentration.
Low PL efficiency.



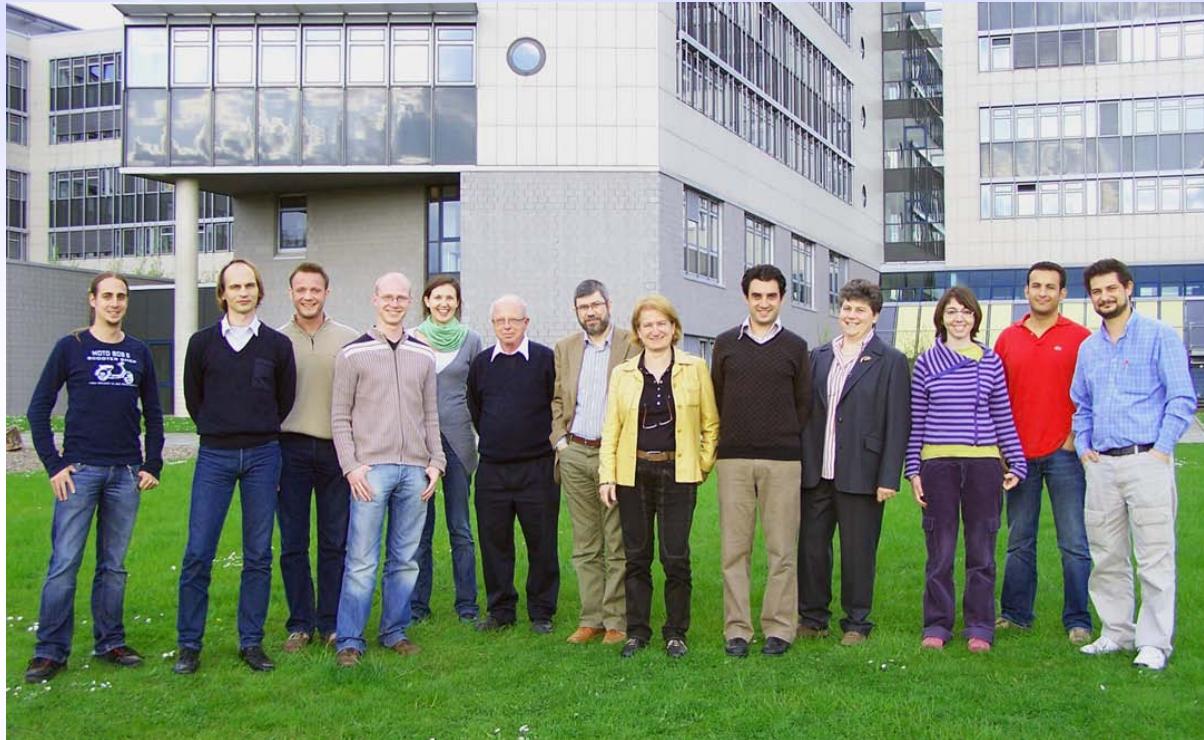
High PL efficiency.

Different N_D and N_{SS} .

Conclusions

- Optical properties and morphology of the nanocolumns are strongly affected by the growth conditions.
- There is electron accumulation at the non-polar surfaces of InN.
- Photoluminescence in InN nanocolumns comes from degenerated electrons recombining with localized holes.
- Differences in the photoluminescence are attributed to different volume and surface charge for each sample. Electron accumulation at non-polar surfaces is not intrinsic.

Acknowledgements:



NanoLICHT European project groups:

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